



## TRIBOLOGICAL PROPERTIES OF SOLID LUBRICANT NANOCOMPOSITE COATINGS OBTAINED BY MAGNETRON SPUTTERED OF $\text{MoS}_2$ /METAL (Ti, Mo) NANOPARTICLES

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In this paper we have investigated the tribological properties of solid lubricant coatings from  $\text{MoS}_2$  nanoparticles. The  $\text{MoS}_2$  were deposited on alloy substrate by magnetron sputtering. The methods were used to create two component coatings in which  $\text{MoS}_2$  were embedded in a Ti matrix. The morphology and microstructure of the coatings were characterized by X-ray diffraction (XRD), scanning electron microscopy (SEM). The tribological properties of  $\text{MoS}_2$ /metal (Ti, Mo) nanocomposite coatings were investigated using a pin-on-disk (POD) tribometer in ambient air and in humidity conditions. A POD test was used to examine the friction behavior and mechanical stability of the coatings, and was carried out in both low and high humidity conditions. XRD and SEM were used to examine morphology of the wear track after the POD test. The  $\text{MoS}_2$  nanoparticles can decrease the friction coefficient of lubricants obviously. However the results showed that their friction reductions have not obvious difference by the POD tribometer. The  $\text{MoS}_2$ /Ti nanocomposite coating showed lower frictions coefficient and higher wear resistance as compared to the pure  $\text{MoS}_2$  coating, which were caused by the microstructure of the composite coating that serve as perfect intermediate lubricants between the contact surfaces. The analyses of surfaces composition coating characterized by XRD and SEM images showed that deposition nanoparticles, form a protective film allowing an increase in the load capacity of friction (rubbed) pairs. The main advantage of the nanoparticles is ascribed to the release and furnishing of the nanoparticles from the valley onto the rubbing metal surface and their confinement at the interface. Coatings of  $\text{MoS}_2$  alone were found to perform well under low-humidity conditions, but poorly under high-humidity. Alloying of  $\text{MoS}_2$  with Ti was found to provide some improvement under high humidity. The patterned film by  $\text{MoS}_2$ /Ti nanocomposite was found with friction and wear properties superior to either Ti or  $\text{MoS}_2$  alone. The effects of metal inclusion in the non-composite were further discussed and compared with the former reports. The coatings investigated here have potential applications for cutting tools and metal forming dies that will enhance tool life and reduce the energy expended due to friction forces, as well as for use in various tribological fields such as seals, bearings or electrical motor brushes and, also, for applications needing excellent lubrication and wear-reducing properties.

*Key words:* lubricant, tribology, nano-composite, coating,  $\text{MoS}_2$  – nanoparticles

### 1. INTRODUCTION

Solid lubricants have the advantages including long life, no contamination and usage in harsh environments that liquid lubricant cannot be used. It has been very well known that  $\text{MoS}_2$ , especially nanosized  $\text{MoS}_2$ , presents considerable applications in many fields such as solid lubrication and for lubricating oils and potential self-lubricating polymer materials [1].

It is reported that  $\text{MoS}_2$  nanoparticles mixed in oil, grease or impregnated into matrix porous of powdered materials appear to enhance the tribological properties in definite loading range in comparison to typical metal dichalcogenides particles [2].

Magnetron sputtering and ion plating technology have been developed to deposit  $\text{MoS}_2$  coatings. There are many different kinds of methods to prepare ultrafine  $\text{MoS}_2$  (molybdenum disulfide) particles. However,

using different preparation methods will result in the variations of morphology and performance of MoS<sub>2</sub> nanoparticles.

When pure MoS<sub>2</sub> coating is exposed to oxygen-rich atmosphere, however, tribological performance deterioration would occur due to the oxidation MoS<sub>2</sub>. It has been found that co-sputtering MoS<sub>2</sub> with metal resulted in better tribological properties than that without sputtering metal. In particular, coatings with Ti inclusion exhibit a stable friction coefficient and better endurance than pure MoS<sub>2</sub> coatings in ambient condition.

During sputtering and rubbing process Ti could prohibit the act of entering of O<sub>2</sub> from destroy the constituent and the structure of the coatings at the same time TiO<sub>2</sub> generated could be used as lubricating material, too.

Manufacturing processes used to cut and shape metals can involve severe high-speed and high-temperature surface interactions between the work piece and cutting tool or die. Coatings for tools and dies can provide reduced friction and improved tool life.

Solid lubricant coatings are attractive because they can reduce friction-generated heat. MoS<sub>2</sub> is a common solid lubricant. However, the use of MoS<sub>2</sub> can be limited by excessive wear, as well as a friction coefficient that is sensitive to humidity [3]. Nonetheless, several studies on solid lubricant coatings demonstrated success in lubricating dry sliding contacts over very long periods in pin-on-disk (POD) or reciprocating sliding experiments [4]. Further studies [3, 4] showed that reservoirs developed where lubricant storage can take place, such as on the ball, around the perimeter of the wear scar, or at the terminal points of a reciprocating wear track.

In this paper, we present tribological properties of the composite coatings and the effects of Ti and Mo inclusion in the composite coatings on their tribological performance in ambient air and in humidity conditions, respectively results on patterned coatings and compared them to alloyed and nanolithic single phase coatings.

## 2. EXPERIMENTS

Films of MoS<sub>2</sub>, Ti and MoS<sub>2</sub>-Ti alloys were deposited by magnetron sputtering. MoS<sub>2</sub>/metal (Ti, Mo) coatings were deposited silicon substrates at room temperature by magnetron sputtering. The MoS<sub>2</sub>/metal ratio in the coatings was controlled by sputtering the composite targets.

Pure Ti target (99,99%) and MoS<sub>2</sub>/metal composite targets (pure MoS<sub>2</sub> mixed with Ti) with a diameter of 10 mm were used.

The composite targets were fabricated by press milling the mixture of pure MoS<sub>2</sub> and metal powder, followed by pressing the mixture under a pressure of 60 MPa. Films were deposited on Si and OLC45 steel substrates.

Pre-sputtering was performed to clean the surface of the targets prior to the depositing process.

For the MoS<sub>2</sub> and MoS<sub>2</sub>-Ti alloy films, a bond layer of approximately 50 nm of Ti was first deposited on the substrate, because it could improve the coating adhesion.

Afterwards, the composite coatings with a thickness 1 μm were obtained by sputtering the composite targets. The content of metal inclusions in the main bulk of the coatings was varied by using different composite targets.

Tribological properties of the composite coatings were performed on a pin-on-disk (POD) tribometer. Table 1 shows the film deposition parameters and compositions.

Table 1. Film deposition parameters and compositions.

Sample	Type	% Mo	% S	% Ti	Mo/S ratio
1	MoS <sub>2</sub>	30.5	69.5	0.0	0.44
2	MoS <sub>2</sub> -Ti alloy	33.1	56.9	10.0	0.58
3	MoS <sub>2</sub> -Ti alloy	24.4	55.4	20.2	0.44
4	MoS <sub>2</sub> -Ti patterned	N/A	N/A	N/A	

The compositions given in Table 1 were established by energy dispersive spectroscopy (EDS) in the SEM. The wear tracks of the coatings were examined by means of SEM.

The tribological properties of deposited films were analyzed using a pin-on-disk (POD) test tribometer.

All tests were carried out using a 10 mm diameter steel pin as the contraface. The tests were run using a load of 1 N and sliding speed of 0.2 m/s, with track diameters between 25 and 35 mm.

The environment was air, but the humidity was set at either a low humidity (LH) range 25-35% r/h or a high humidity (HH) range of 65-75% r/h. During the test the sliding load was monitored and recorded every 5 seconds in order to determine the friction coefficient vs number of cycles. After the test, the sample was examined using both X-ray diffraction (XRD) and scanning electron microscopy (SEM).

### 3. RESULTS AND DISCUSSION

The film compositions analyzed by EDS analysis are shown in Table 1, along with the Mo/S ratio for all films. With exception of a sample 2 the films appear substoichiometric with respect to S content.

The patterned film (4 in Table 1) was examined in the SEM, and a cross-section of the as-deposited film on Si is shown in Figure 1. X-ray diffraction (XRD) was also carried out for these films, and the results showed the films were mostly amorphous.

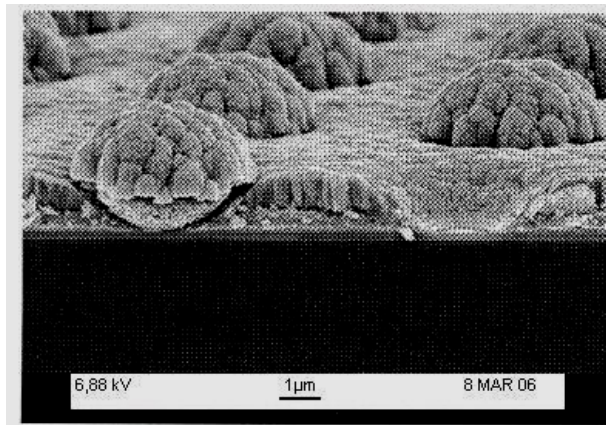


Figure 1. Cross-section SEM image of sample MoS<sub>2</sub>-Ti patterned.

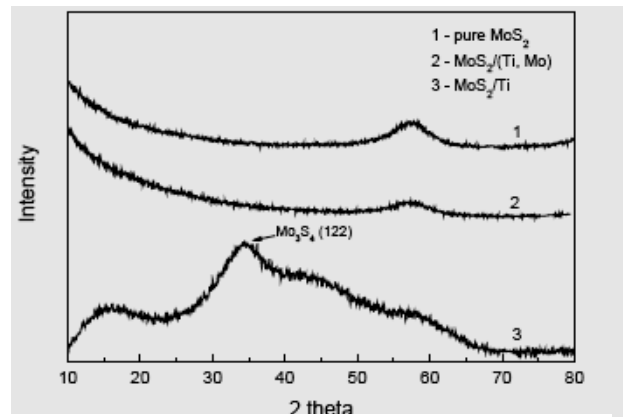


Figure 2. X-ray diffraction patterns of the composite coatings deposited with different targets.

The rounded mounds are MoS<sub>2</sub> and the adjacent film is a bilayer of Ti (lower) and MoS<sub>2</sub> (upper section). The scale marker is equal 1 μm.

Figure 2 shows the XRD patterns of deposited coatings on silicon (Si) substrate. From the XRD analysis, it can be observed that the deposition coatings have a disorderly structure. No peaks of crystalline phase are found from the pure MoS<sub>2</sub> and MoS<sub>2</sub>/(Ti, Mo) coating.

In contrast to the pure MoS<sub>2</sub> and MoS<sub>2</sub>/(Ti, Mo) coatings, an apparent peak of Mo<sub>3</sub>S<sub>4</sub> appears in pattern 3 for the MoS<sub>2</sub>/Ti coating, which is in agreement with the result of addition of Ti to the coatings inhibiting the formation of crystalline MoS<sub>2</sub>.

In this present paper, the sufficient addition of Ti prohibited the crystalline of MoS<sub>2</sub> and promoted the formation of Mo<sub>3</sub>S<sub>4</sub>.

Relative sputtering without the addition of metal was difficult to produce the crystalline MoS<sub>2</sub>.

The variations of friction coefficient vs. number cycles functioning of the coatings in ambient air at a sliding speed of 0.2 m/s and an applied load of 1N are shown in Figure 3. After running-in, the coatings show the steady friction state and long wear endurance. The pure MoS<sub>2</sub> coating demonstrated a friction coefficient of 0.05 and did not fail until 36000 cycles.

The addition of Ti in MoS<sub>2</sub> coating decreased the friction coefficient, as presented in curve (3), consequently, the steady friction coefficient reached about 0.03. However, the addition of Mo in the composite coatings increased the friction coefficient to about 0.08 when rubbing state was stabilized. Under lower applied load as our paper, Ti assisted the sliding of the MoS<sub>2</sub> coating more easily.

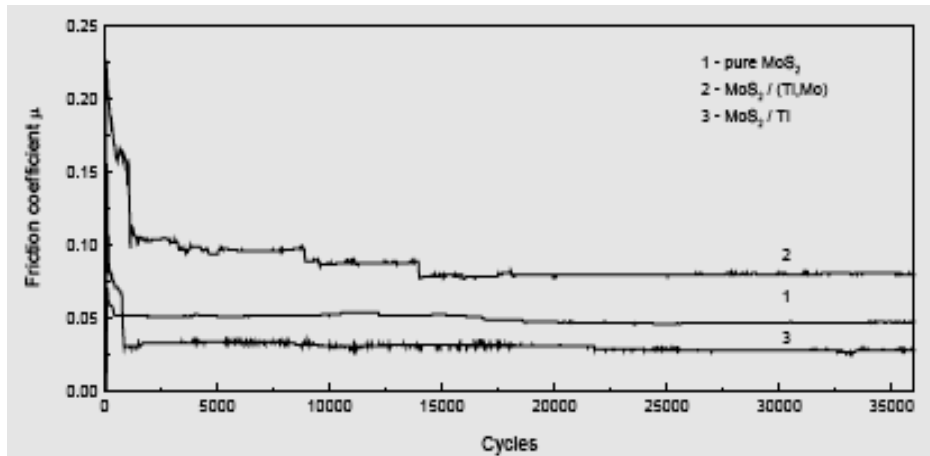


Figure 3. SEM micrographs of the surface morphology of the coatings: 1 - pure MoS<sub>2</sub>; 2 - MoS<sub>2</sub>/(Ti, Mo); 3 - MoS<sub>2</sub>/Ti.

Figure 4 shows the wear tracks of the coatings after microfriction tests. No spalling of the coatings were observed, which was coincided with the steady friction coefficient until the sliding end of 36000 cycles.

In comparison with the wear tracks of the coatings shown in Figure 4 a, b, c, it can be found that morphology of rubbing surface of MoS<sub>2</sub> /Ti coating is smoother than the other two coatings. In the contrast to the MoS<sub>2</sub> /Ti coating, the obvious scratch tracks on the MoS<sub>2</sub> /(Ti, Mo) coating are observed, as shown in Figure 4b.

Lots of wear scar are presented on the rubbing surface. As indicated in XRD pattern (see Figure 1), the MoS<sub>2</sub> /(Ti, Mo) coating has poor crystalline MoS<sub>2</sub> structure. Otherwise, MoS<sub>2</sub>/Ti coating has not the distinguished MoS<sub>2</sub> structure, the main structure is Mo<sub>3</sub>S<sub>4</sub>.

Therefore, simultaneous addition of Mo and Ti increase the formation of the MoS<sub>2</sub>, which is adverse for the friction coefficient and wear resistance proved by our experiments. While sufficient addition of Ti alone would facilitate the formation of Mo<sub>3</sub>S<sub>4</sub>, Mo<sub>3</sub>S<sub>4</sub> resulted from addition of Ti improved the tribological properties, also, the formation of TiO<sub>2</sub> in the coating as a result of oxygen gettinger during the rubbing test could facilitate superior friction coefficient and wear resistance.

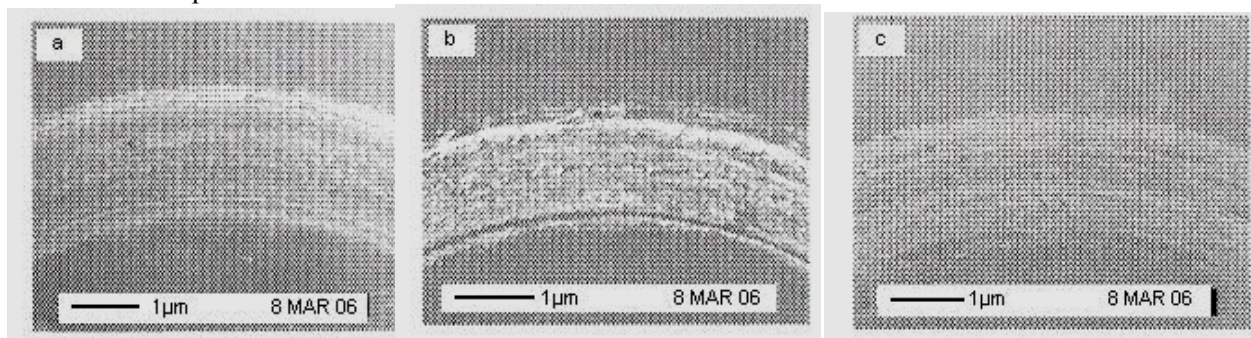


Figure 4. SEM micrographs of the wear tracks of the coatings: a - pure MoS<sub>2</sub>; b - MoS<sub>2</sub>/(Ti, Mo); c - MoS<sub>2</sub>/Ti.

Tribological testing of the MoS<sub>2</sub> coatings was carried out under dry and humid conditions. For the dry condition, the friction coefficient remained at 0.08 for 8000 cycles, at which point the test was terminated (see Fig. 5). Under humid conditions, the friction coefficient started at 0.2 and rose to 0.5 after 5000 cycles.

This test confirmed the well-known effect of humidity through the friction behavior of MoS<sub>2</sub> coatings [3].

The friction behavior of the MoS<sub>2</sub>-Ti alloy coatings were also tested under both low and high humidity conditions.

Under low humidity conditions sample 2 (see Table 1) maintained a low friction coefficient (0.1) while sample 3 was slightly heigher in friction (0.18). Under high humidity conditions, the coating of sample 1 (MoS<sub>2</sub>-Ti alloy) has a reasonable friction coefficient (average near 0.2), but exhibits a high scattering

compared to the result for the same coating under low humidity conditions. Sample 3 (see Table 1) appeared worse and the test was terminated after 3500 cycles. Overall, the addition of a smaller amount of Ti (10 %) improves the friction behavior under high humidity conditions, but does not completely cancel the effects of humidity.

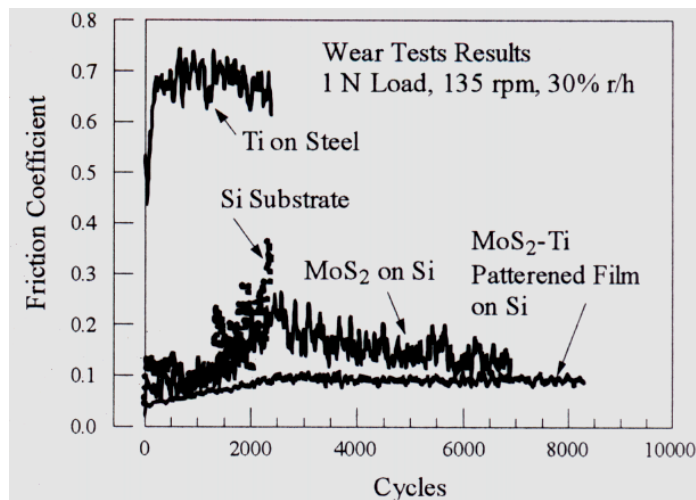


Fig. 5. Pin-on-disk test result for the patterned MoS<sub>2</sub>-Ti coating and comparison to MoS<sub>2</sub>, Si and Ti.

Fig. 5 shows the results for the patterned MoS<sub>2</sub>-Ti coating (on Si substrate) and others under low humidity conditions.

The patterned coating maintained a low friction coefficient of 0.1 for the duration of the 8000 cycles test. The test result for the Si substrate is shown, and a similar result is obtained for MoS<sub>2</sub> on Si, which indicates the coating had worn through the Si substrate. A coating of Ti on steel is also shown, and this coating exhibited a very short life. Therefore, for coatings on Si, the patterned MoS<sub>2</sub>-Ti coating gave a longer wear life and a lower friction coefficient than those of the constituent (Ti or MoS<sub>2</sub>) coatings alone.

#### 4. CONCLUSIONS

Addition of Ti helps the formation of Mo<sub>3</sub>S<sub>4</sub>, which is favourable to the decrease of the friction coefficient. The MoS<sub>2</sub>/Ti composite coating showed the superior friction coefficient and wear resistance in ambient air.

Except Ti, the addition of Mo helped the formation of MoS<sub>2</sub> structure. The bigger flakes in the MoS<sub>2</sub> (Ti, Mo) composite coatings were not favourable to the improving tribological properties as compared to the coatings without metal inclusions.

Effect of humidity through the friction behavior of MoS<sub>2</sub> coatings is different as a function of humidity conditions (low or high humidity) and of the Ti percent.

The coatings on Si, the patterned MoS<sub>2</sub>-Ti coating gave friction and wear properties superior (a longer wear life and a lower friction coefficient) than those of the constituent (Ti or MoS<sub>2</sub>) coatings alone.

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